

1 **Aerogel sponges of silk fibroin, hyaluronic acid and heparin for soft tissue**
2 **engineering: Composition-properties relationship**

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10 #Equivalent contribution

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14

15 **ABSTRACT**

16 This work aims to design biocompatible aerogel sponges that can host and control the release of
17 stromal cell-derived factor-1 α (SDF-1 α or CXCL12), a key protein for applications ranging from
18 regenerative medicine to cancer therapy (notably for neural tissues). Miscibility of silk fibroin (SF)
19 and hyaluronic acid (HA) was investigated by means of fluorescence and scanning electron
20 microscopy to identify processing conditions. Series of freeze-dried sponges were prepared by
21 associating and cross-linking within the same 3D structure, HA, SF, poly-L-lysine (PLL) and heparin
22 (hep). Aerogel sponges presented high swelling degree and porosity (~90%), adequate mean pore
23 diameter (ca. 60 μ m) and connectivity for welcoming cells, and a soft texture close to that of the
24 brain (6-13 kPa Young's Modulus). Addition of SF yielded sponges with slower biodegradation. SF-HA
25 and SF-HA-hep sponges retained 75% and 93% of the SDF-1 α respectively after 7 days and were
26 found to be cytocompatible *in vitro*.

27

28 *Keywords:* Brain biomimicry; SDF-1 α controlled release; porous scaffold; aerogel sponges; silk fibroin.

29

30 1. Introduction

31 Efficient three-dimensional (3D) scaffolds serving as cell supports and tissue bridges demand
32 composition and properties tailored to each specific target tissue and purpose. In the case of neural
33 tissue, brain mimicking may help improving scaffolds for not only regenerative purposes (Boni, Ali,
34 Shavandi, & Clarkson, 2018; Dehqan Niri, Karimi Zarchi, Ghadiri Harati, Salimi, & Mujokoro, 2019;
35 Orive, Anitua, Pedraz, & Emerich, 2009) but also for localized treatments, such as brain cancer
36 management through environmental traps that control the progression of the disease (Haji Mansor
37 et al., 2018; Najberg, Haji Mansor, Boury, Alvarez-Lorenzo, & Garcion, 2019). Brain damage might
38 cause cognitive, behavioral and physical disabilities. Depending on the cause, treatments exist to
39 control the damages or slow down the deterioration of the brain. Unfortunately, there is none
40 reconstructive intervention available and the central nervous system (CNS) has limited capacity to
41 spontaneously regenerate. It is therefore necessary to develop innovative strategies to promote
42 tissue repair. One strategy consists in the incorporation to a scaffold of molecules that can attract
43 and support integration of endogenous neural progenitor/stem-cells (NPSCs) to the site of
44 implantation. SDF-1 α , also known as CXCL12, is a chemokine that has been pointed out as a potent
45 candidate to improve outcomes in brain injuries due to its well-described involvement in NPSCs
46 homing (Christie & Turnley, 2013; Imitola et al., 2004). However SDF-1 α bolus administration in the
47 brain leads to poor results in the long term because of its rapid clearance (Dutta et al., 2017).
48 Encapsulating this chemokine in nanoparticles or immobilizing it in scaffolds may prevent premature
49 degradation and may provide sustained release for NPSCs attraction (Addington et al., 2015; Dutta,
50 Fauer, Mulleneux, & Stabenfeldt, 2015; Jian et al., 2018). SDF-1 α has also been shown to accelerate
51 the healing in a variety of other tissues, **but prolonged levels of this chemokine are not easy to**
52 **provide** (Hu et al., 2018; Lau & Wang, 2011; Rabbany et al., 2010; Zhao, Jin, Li, Qiu, & Li, 2017).

53 The 3D scaffold itself should be adequately designed in terms of architecture and physicochemical
54 properties. The brain is made up of 80% water and has a Young's modulus of less than 1 kPa (De
55 Souza & Dobbing, 1971; Engler, Sen, Sweeney, & Discher, 2006; Nava, Raimondi, & Pietrabissa,
56 2012). Aerogel sponges that can transform into soft hydrogels once hydrated are good candidates to
57 mimic these features; indeed, hydrogel-like bioscaffolds have been pointed out as ideal ones for
58 brain regeneration (Modo, 2019). The scaffold should be designed with pores that are small enough
59 to support 3D cell-cell contacts and large enough to allow good diffusion of nutrients, oxygen and
60 bioactive factors for cell survival and growth. Pore sizes between 20 and 160 μm have been identified
61 as adequate for cell growth and colonization (Mahumane, Kumar, Du Toit, Choonara, & Pillay, 2018).
62 In this context, freeze-dried sponges are promising scaffolds for soft tissue regeneration including the
63 neural one as they can mimic soft tissues peculiarities (Führmann, Obermeyer, Tator, & Shoichet,

64 2015; Haring et al., 2019; Seyedhassantehrani, Li, & Yao, 2016). Specifically, compared to more
65 compact sponges, aerogel-based nanostructured sponges are characterized by their lightweight and
66 interconnected open porosity, with a texture particularly suitable for cell attachment (Garcia-
67 Gonzalez et al., 2019).

68 HA is one of the main constituents of the brain extracellular matrix (ECM) (Roulahti, 1996). It is
69 known to enhance cell mobility and invasion by interacting with the CD44 receptor (Bellail, Hunter,
70 Brat, Tan, & Van Meir, 2004). The main concern with HA as a scaffold is its quick degradation *in vivo*.
71 Differently, SF from the silk worm *Bombyx mori* is known for its biocompatibility, relatively slow
72 degradability and versatile processing abilities (Mobini et al., 2016; Thurber, Omenetto, & Kaplan,
73 2015; Vepari & Kaplan, 2012). Based on its unique mechanical properties that feature good strength
74 and elasticity, it is possible to produce hydrogels, sponges, films and nanofibers via several organic
75 solvent-free processes (Rockwood et al., 2011). Relevantly, SF has shown low immunogenicity and
76 excellent compatibility with brain, promoting reparative mechanisms (González-Nieto, Fernández-
77 García, Pérez-Rigueiro, Guinea, & Panetsos, 2018; Martín-Martín et al., 2019). Like HA, heparin (hep)
78 is another component of the ECM present in the brain. Due to its strong capacity to bind
79 chemokines, the incorporation of this molecule in a scaffold has been shown to facilitate the loading
80 of SDF-1 α (Krieger et al., 2016; Prokoph et al., 2012).

81 The present work relies on the hypothesis of that suitable combinations of HA, SF and hep may
82 render 3D aerogel scaffolds with synergic features in terms cell infiltration (HA), textural and
83 mechanical properties (SF) and control of SDF-1 α uptake and release (hep) avoiding premature
84 clearance. The aim is to obtain mid-term stable scaffolds that can attract cells through the creation of
85 a SDF-1 α gradient. To carry out the work, miscibility of HA with SF was first evaluated under different
86 processing conditions due to the reported phase-separation problems found between SF and
87 polysaccharides (Chen, Li, & Yu, 1997; Garcia-Fuentes, Giger, Meinel, & Merkle, 2008). Then, sponges
88 were prepared by the crosslinking of the polymers with N-(3-dimethylaminopropyl)-N'-
89 ethylcarbodiimide hydrochloride (EDC) and N-hydroxysulfosuccinimide sodium salt (NHS) followed by
90 freeze-drying. Aerogel sponges architecture and physicochemical characteristics (porosity,
91 degradation, water uptake and stiffness) were measured as well as their cytocompatibility and *in*
92 *vitro* release profile of SDF-1 α under *in vivo* mimicking conditions.

93 **2. Materials and methods**

94 **2.1. Materials**

95 Hyaluronic acid (HA) with a mean molecular weight of 360 kDa (200-400 kDa range), glucuronic acid
96 47.4% and intrinsic viscosity 7.7 dL/g was supplied by Guinama (Valencia, Spain). Poly-L-lysine (PLL)

97 with a molecular weight of 150-300 kDa (0.1% w/v in H₂O) was purchased from Sigma-Aldrich (St
98 Louis, MO, USA). Heparin sodium salt (hep) from porcine intestinal mucosa with a molecular weight
99 of 15,000 ± 2,000 Da was purchased from Calbiochem (Billerica MA, USA). HEPES buffer, bovine
100 serum albumin (BSA), resazurin, hyaluronidase from bovine testes type I-S, heparinase I and III blend
101 from *Flavobacterium heparinum* (50 U) and sodium chloride were purchased from Sigma-Aldrich (St
102 Louis, MO, USA). EDC and NHS were from Acros Organics (New Jersey, USA). Human SDF-1 α labelled
103 with AlexaFluor647 (AF-SDF-1 α) was purchased from Almac Sciences (Craigavon, Northern Ireland).
104 Human SDF-1 α was purchased from Miltenyi Biotec (Paris, France).

105 **2.2. Silk fibroin solution preparation and characterization**

106 Solution of SF in water was produced by IMIDA (Murcia, Spain) under GMP conditions. Cocoons of
107 the silkworms *Bombix mori* (reared in IMIDA, Murcia, Spain) were chopped into 4 pieces then boiled
108 in a 0.02 M Na₂CO₃ aqueous solution for 30 minutes to remove the sericin protein. Fibers were rinsed
109 3 times in distilled water and dried at RT for 3 days. They were dissolved in a 9.3 M LiBr solution for
110 3h at 60°C to generate a 20% (w/v) solution which was then dialyzed against distilled water for 3 days
111 at 4°C (Snakeskin Dialysis Tubing 3.5 KDa MWCO, Thermo Scientific). The obtained 8% (w/v) SF
112 solution in water was stored at 4°C for less than 1 month.

113 SF was characterized using SDS-poly(acrylamide) gel electrophoresis (SDS-PAGE) as follows. A SF
114 stock solution at 10 mg/mL was prepared in water and diluted in 5x Laemmli buffer (90 mmol of
115 0.5M Tris-HCl, pH 6.8, 20% glycerol, 4% SDS, 0.004% bromphenol blue and 10% β -mercaptoethanol)
116 to obtain a final concentration of 6.25 μ g/ μ L and boiled at 95°C for 5 min. An aliquot (10 μ l) was
117 added into the wells. Electrophoresis was performed using a 4-20% gradient mini-protean TGX
118 precast gel (Bio Rad). HiMark Pre-stained protein standards (Thermo Fisher) served as molecular
119 weight marker. After electrophoresis, the gel was stained with solution 1% (w/v) Coomassie Brilliant
120 Blue R- 250, 10% (v/v) acetic acid, 40% (v/v) methanol and distilled water over night, and destained
121 with 10% (v/v) acetic acid, 40% (v/v) methanol and distilled water for 5h. Three different batches
122 were analyzed.

123 **2.2. Films preparation and characterization**

124 Dispersions of HA 4% (w/v), SF 4% (w/v) or SF 4% (w/v) with HA 2% (w/v) in HEPES buffer ($C_{\text{HEPES}} =$
125 $20 \cdot 10^{-3}$ M, $C_{\text{NaCl}} = 0.15$ M, pH = 7.4) were prepared under magnetic stirring for either 15 min, 2 h or 15
126 h. Films were prepared by pouring the dispersions in Petri dishes and left to dry **24 h** at RT. Films
127 morphology was observed by scanning electron microscopy (SEM) with a SEM Evo LS15 (Zeiss, USA)
128 after being coated with iridium.

129 **2.3. Aerogel sponges preparation and characterization**

130 **2.3.1 Aerogels preparation**

131 HA was dissolved in HEPES buffer ($C_{\text{HEPES}} = 20 \cdot 10^{-3}$ M, $C_{\text{NaCl}} = 0.15$ M, pH=7.4) to obtain a final
132 concentration of 4% (w/v). Sponges were prepared with 4 different compositions by first mixing the
133 polymers together: HA 4% (w/v) with PLL 0.01% (w/v) (HA-PLL), SF 4% (w/v) (SF), SF 4% (w/v) with HA
134 2% (w/v) (SF-HA), and SF 4% (w/v) with HA 2% (w/v) and hep 1% (w/v) (SF-HA-hep). For the
135 preparation of the SF-HA and the SF-HA-hep sponges, SF 8% (w/v) solution was gently stirred 5h with
136 an equivalent volume of HA 4% (w/v) solution in HEPES buffer with or without heparin sodium salt.
137 EDC and NHS were dissolved in H₂O ($C_{\text{EDC}} = 50$ mg/mL, $C_{\text{NHS}} = 18$ mg/mL) and then immediately added
138 to the HA-PLL or SF-HA solution with a 1:10 (crosslinkers/polymers, v/v) ratio and mixed rapidly. For
139 the SF-HA-hep, a higher concentration of crosslinker was needed ($C_{\text{EDC}} = 150$ mg/mL, $C_{\text{NHS}} = 55$
140 mg/mL). The solutions were poured immediately in a 96-well plate, covered and allowed to crosslink
141 for 15 h at 4°C. The gels were then frozen at -20°C for 24 h, and freeze-dried in a Telstar® LyoQuest at
142 -70°C and 0.01 mBar overnight. After freeze-drying, the aerogel sponges were removed from the
143 wells and stored in a desiccator with silica gel. For aerogels containing SF, an annealing step was
144 applied to increase the crystallinity of the protein in order to obtain sponges that maintain their
145 structural integrity in aqueous solutions (Hu et al., 2011). Annealing with ethanol vapor consisted in
146 placing the sponges in a desiccator at atmospheric pressure with ethanol 70% (v/v) at the bottom
147 and kept for 2 h at RT. The sponges were then freeze-dried again to remove traces of ethanol.

148 **2.3.2 Fluorescence microscopy of internal structure of the aerogel sponges**

149 Sponges were frozen in liquid nitrogen and immobilized in Tissue-Tek® (Sakura, USA). Cross sections
150 with 30 μm thickness were obtained using a cryostat® (Leica CM3050 S Research Cryostat, Leica
151 Biosystems Nussloch GmbH, Germany). Sections were placed on Starfrost® slides coated with gelatin
152 and chrome alum, incubated 5 min with ActinRed (ThermoFisher), and the lamella mounted on the
153 sections with fluorescence mounting medium (Dako, Agilent Technologies, USA). Finally, the slide
154 was observed under a fluorescence microscope (Carl Zeiss Axioskop 2 MOT, Germany) with a
155 rhodamine filter, and pictures were taken with the same exposition time of 58 ms between slides.

156 **2.3.3 FTIR analysis**

157 Attenuated total reflection Fourier transformed infra-red (ATR-FTIR) analysis of sponges (10 mg) was
158 performed with a Gladi-ATR equipped with a diamond crystal (Pike, Madison, WI, USA). For each
159 measurement, 64 scans were recorded with a resolution of 4 cm⁻¹, and the wavelength ranged from
160 400 to 4000 cm⁻¹. The increase in crystallinity of SF sponges after ethanol annealing was evaluated in
161 terms of percentage of β-sheets applying the method described by Hu *et al.* (Hu, Kaplan, & Cebe,

162 2006). Briefly, Fourier self-deconvolution (FSD) of the IR spectra covering the amide I region (1595-
163 1705 cm^{-1}) was performed using Opus 7.8 software (noise reduction factor of 0.3 and bandwidth of
164 2.5) followed by a curve fitting on this region. Peaks at 1616-1621, 1622-1627, 1628-1637, and 1697-
165 1703 corresponded to β -sheets (Hu et al., 2006).

166 2.3.4 Porosity evaluation

167 Density and porosity (P) of aerogel sponges was calculated using Eq. (1) and Eq. (2). The thickness
168 and diameter of the specimens were measured using a digital caliper. Polymer density was supposed
169 to be equal to 1. For each composition, three sponges were measured.

$$170 \text{ Sponge density (g.cm}^{-3}\text{)} = \frac{\text{Sponge mass (mg)}}{\text{Sponge volume (mm}^3\text{)}} \quad \text{Eq. (1)}$$

$$171 P (\%) = \left(1 - \frac{\text{Sponge density (g.cm}^{-3}\text{)}}{\text{Polymer density (g.cm}^{-3}\text{)}} \right) \times 100 \quad \text{Eq. (2)}$$

170

171 2.3.5 Water uptake

172 The capability of the sponges to uptake water was evaluated, in triplicate, after immersion at RT in 3
173 ml of NaCl 0.9%. The sponges were withdrawn using tweezers, gently treated with blotting paper for
174 10 s to remove the unabsorbed excess of medium, weighed, and immediately returned to the
175 corresponding vial. The weight measurements were repeated until the hydration equilibrium was
176 reached. Water uptake was calculated as follows:

$$177 \text{ Water uptake (\%)} = \left(\frac{\text{hydrated sponge (mg)} - \text{dry sponge (mg)}}{\text{hydrate sponge (mg)}} \right) \times 100 \quad \text{Eq. (3)}$$

178

179 2.3.6 Young's modulus measurement

180 Hydrated sponges were cut to obtain cylinders with a thickness of about 3 mm. The cylinders were
181 placed on the platform of a texture analyzer (TA.TX Plus, Texture Technologies, USA) and compressed
182 downward up to 1.5 mm with a speed of 1 mm/s. Three hydrated sponges were analyzed for each
183 composition. The Young's modulus was determined from the slope of the curve obtained by plotting
184 the force (F) needed to compress the sponge divided by the cross-sectional area of the sponge (S)
185 against the deformation of the sponge (ΔL) divided by the initial length of the sponge (L) (Lassoued,
186 Delarue, Launay, & Michon, 2008) as follows

$$187 E \text{ (MPa)} = \frac{F \text{ (N)} / S \text{ (mm}^2\text{)}}{\Delta L \text{ (mm)} / L \text{ (mm)}} \quad \text{Eq. (4)}$$

188 **2.3.7 Degradation**

189 *In vitro* erosion of sponges was monitored, in triplicate, in PBS with enzymes (0.3 mg/mL
190 hyaluronidase (Correia et al., 2011) and 0.1 U/mL heparinase) at 37°C for 7 and 21 days. Aerogel
191 sponges were cut to have similar sizes (6 mm diameter and 3 mm height) and weighed. After 7 or 21
192 days in 3 mL medium, the sponges were washed with water, snapped frozen in liquid nitrogen and
193 then freeze-dried. Sponge degradation was calculated as percent of mass loss (aerogel sponge mass-
194 degraded sponge mass) referred to the original aerogel sponge mass.

195 **2.3.8 Scanning electron microscopy (SEM)**

196 Sponge morphology (surface and transversal cut) was observed using a SEM Evo LS15 (Zeiss, USA).
197 Sponges were firstly hydrated in NaCl 0.9% (w/v) aqueous solution then, either directly or after 7
198 days at 37°C, snapped frozen in liquid nitrogen. Pore diameter was measured using ImageJ 1.50
199 software; at least 94 pores were measured per sponge.

200 **2.4. Cytocompatibility**

201 The cytocompatibility was evaluated using human fibroblast cells (NIH/3T3, ATCC® CRL-1658™) that
202 were cultured at 37°C and 5% CO₂ in DMEM high glucose media (Sigma Aldrich) supplemented with
203 10% fetal bovine serum, and 1% penicillin/streptomycin and subcultured every 3-4 days. Cells were
204 seeded in a 24-well plate (20,000 cells/well) for 24 h. Hydrated HA-PLL, SF, SF-HA and SF-HA-hep
205 sponges in PBS were firstly cut to obtain cylinders with 2 mm height and then sterilized under UV
206 light for 1 h. Immediately before use, sponges were washed 3 times with PBS under sterile conditions
207 and then equilibrated in complete DMEM for 15 min. Suspended culture inserts (MilliCell, PET, 8 µm)
208 were placed in the wells with sponges inside in triplicate. Inserts without sponges were used as
209 control. After 24 h of incubation at 37°C and 5% CO₂, inserts and sponges were removed, and the
210 media was replaced with 500 µL of resazurin 44 µM. After 2 h of incubation at 37°C, cell viability was
211 estimated from the fluorescence intensity of the reduced product of resazurin, called resorufin,
212 which is formed in the cytosol of viable cells. Fluorescence was measured using a ClarioStar
213 microplate fluorometer (BMG Labtech GmbH, Ortenberg, Germany) at $\lambda_{\text{ex}} = 545 \text{ nm}$ and $\lambda_{\text{em}} = 600$
214 nm. All readings were normalized to those obtained with the control wells.

215 **2.5. Analysis of SDF-1 α uptake and release**

216 **2.5.1. Distribution of SDF-1 α in the sponges**

217 SF-HA and SF-HA-hep sponges were washed in PBS (as in 2.4) and cut to obtain cylinders of 2 mm
218 height and 3 mm diameter. Excess of PBS was removed with Nunc paper. A drop of 3 µL of AF-SDF-1 α
219 (300 ng in PBS) was added on top of the sponges, which were then incubated 1h at 4°C. A map of the

220 fluorescence of the sponges ($\lambda_{\text{ex}} = 625 \text{ nm}$; $\lambda_{\text{em}} = 680 \text{ nm}$) was performed with a spectrophotometer
221 (CLARIOstar®plus, BMG Labtech, Orthenberg, Germany). Each point was measured 500 μm apart.

222 **2.5.2. Quantitative analysis of SDF-1 α**

223 SF-HA and SF-HA-hep sponges were hydrated and cut as above (section 2.5.1). Then, the sponges
224 were placed in Eppendorf tubes coated beforehand with Sigmacote® (Sigma-Aldrich, USA) and
225 sterilized 1h with UV light before adding 3 μL of SDF-1 α (300 ng in PBS) on top of the sponges. The
226 sponges were incubated for either 1 h or 12 h at 4°C. PBS (500 μL) with 0.1% BSA and enzymes (0.3
227 mg/mL hyaluronidase and 0.1 or 0.5 U/mL heparinase) was added in each tube and incubated at
228 37°C under gentle oscillation. At least three sponges of each composition were tested for each
229 different loading/release condition. The release medium was collected at defined time points (1 h, 8
230 h, 1 d, 2 d, 3 d and 7 d) and an equal volume of fresh medium was added back to the sponges. All
231 solutions were collected and frozen at -20°C until ELISA tests (R&D Systems, Lille, France)

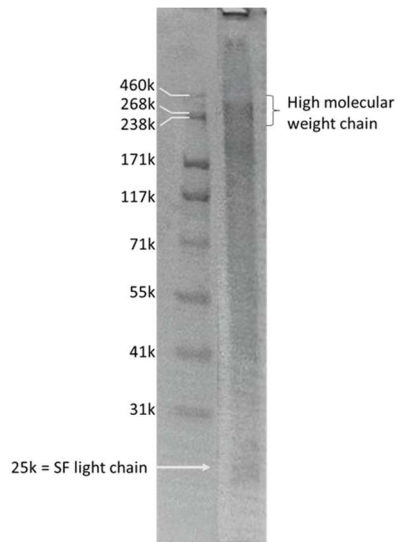
232 **2.6. Statistical analysis**

233 The data were statistically analyzed using an ANOVA test with Prism 7 software. In all statistical
234 comparisons $p < 0.05$ was considered statistically significant.

235 **3. Results and Discussion**

236 **3.1. Silk fibroin characterization**

237 In the gland of *Bombix mori*, SF is composed of heavy molecular weight chains (H-chains) of about
238 370 kDa) and light molecular weight chains (L-chains) of about 25 kDa (Yamada, Nakao, Takasu, &
239 Tsubouchi, 2001). Depending on the process used to obtain the SF solution, SF can be more or less
240 degraded, which may have an impact on the properties of the materials produced (Nultsch &
241 Germershaus, 2017). The molecular weight profile of the SF obtained under GMP conditions by
242 IMIDA was evaluated by means of SDS-PAGE with a 4-20% gradient gel (Figure 1). The L-chain at 25
243 kDa was clearly observed. The H-chain was recorded between 238 kDa and 460 kDa. These results
244 confirm that the process used to produce the SF solution was able to retain the integrity of the
245 protein.

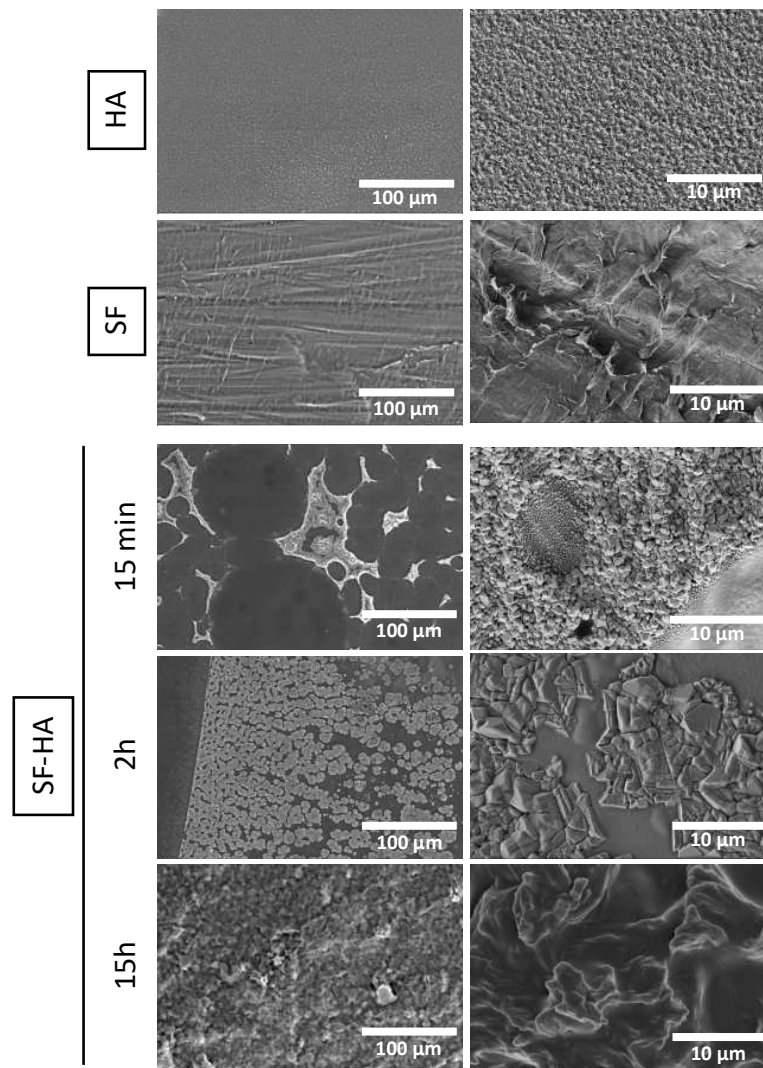


246

247 Figure 1: Molecular weight analysis of the SF solution by SDS-PAGE showing the expected difference
 248 in electrophoretic mobility between the high molecular weight protein and the low molecular weight
 249 protein.

250 3.2. Kinetic study of the interaction of SF with HA

251 Homogeneous dispersions of SF with hydrophilic polymers such as HA have been previously shown to
 252 be quite difficult to attain. Phase separation and/or specific interactions are quite prone to occur
 253 (Garcia-Fuentes et al., 2008; Garcia-Fuentes, Meinel, Hilbe, Meinel, & Merkle, 2009). Thus, the effect
 254 of the mixing time on the inner structure of SF-HA systems was investigated by preparing films
 255 composed of either HA, SF or a blend of SF and HA that had been stirred for 15 min, 2 h or 15 h
 256 (Figure 2). The HA film displayed a rough surface while the SF film was reproducing the petri dish
 257 pattern. When mixed together for 15 min, two separate phases appeared in the form of globules
 258 with sizes ranging from 9.6 to 128.2 μm (38.7 ± 23.5 in average) that were clearly outlined by the
 259 second phase. The phase inside the globules had a similar topography to that of the SF film while the
 260 other phase appeared to be similar to that of the HA film. After 2 h stirring, the globules became
 261 smaller (2.4 – 19 μm), resulting in an increase in homogeneity. After overnight mixing, the film
 262 adopted a much more homogeneous appearance. However, large aggregates could be seen to form
 263 extensively across the surface of the film. Therefore, an intermediate mixing time of about 5h for
 264 applied for sponge preparation.



265

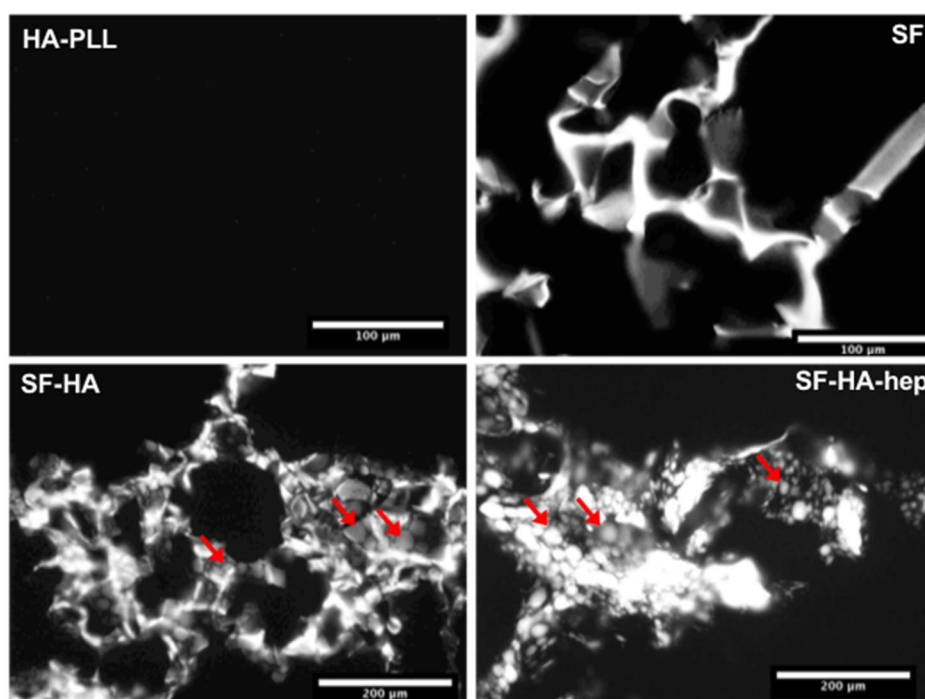
266 Figure 2: SEM images of films prepared with either HA, SF or a blend of SF and HA mixed for 15 min,
 267 2h or 15h.

268 3.3. Structural characterization of the aerogel sponges

269 HA-PLL, SF, SF-HA and SF-HA-hep aerogel sponges were prepared by crosslinking of the polymers
 270 followed by freeze-drying. A mixture of HA and PLL (only 0.01% w/v) was used instead of HA solely
 271 since the reaction with EDC and NHS requires a carboxylate and a primary amine (Kuo, Swarm, &
 272 Prestwich, 1991). Carboxylate moieties are present on the HA chains but not the amine groups.
 273 Therefore, this crosslinking reaction does not proceed well for HA solely. It is known that PLL
 274 facilitates cellular adhesion via electrostatic interactions and may improve HA performance in this
 275 regard (Mazia, Schatten, & Sale, 1975), but the proportion of PLL was kept low to facilitate
 276 comparisons with the other sponges.

277 The microstructure of the aerogel sponges' walls was observed under fluorescence microscopy
 278 (Figure 3). Antibodies used in immunofluorescence imaging like DAPI or ActinRed fixed in a non-

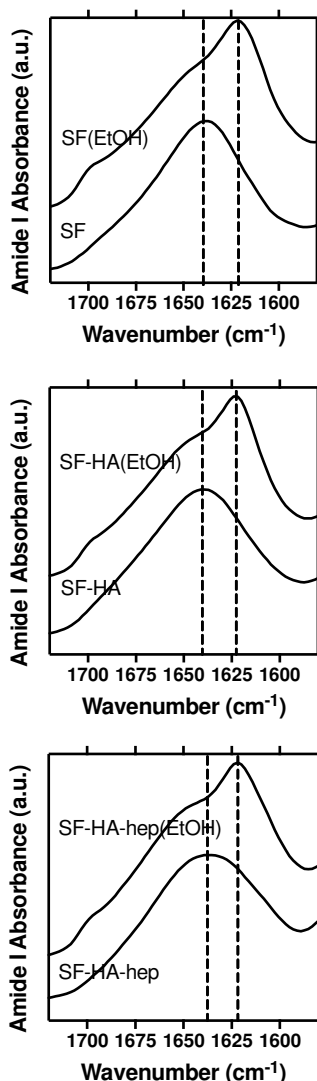
279 specific manner to SF without interacting with HA. Indeed, a much stronger signal was observed for
280 the SF sponge compared with the HA-PLL sponge. The walls of the SF sponge showed a smooth
281 surface. Upon the addition of HA, the walls of the pores of SF-HA sponge became irregular and
282 seemed to be formed by small fluorescent aggregates. Such aggregates were intensified upon the
283 introduction of hep. The structure of SF-HA and SF-HA-hep aerogel sponges was formed by
284 fluorescent aggregates that should therefore be SF-rich domains distributed in a non-fluorescent
285 matrix composed mainly of HA. Garcia-Fuentes *et al.* (2008) have previously reported that SF/HA
286 mixtures at 80:20 and 60:40 weight ratios may develop phase separation, with one phase containing
287 HA alone while the other phase consisted of a mixture of HA and SF. The phase separation observed
288 in the films prepared by us (SF/HA 2/1 ratio) as well as the structure observed in the pore walls of the
289 aerogel sponges are consistent with those previous findings. We can therefore pointed out that the
290 polymer ratio as well as the mixing time are critical parameters when SF and HA are used to prepare
291 3D scaffolds.



292
293 Figure 3: Autofluorescence of the aerogel sponges under Rhodamine filter with the same exposure
294 time of 58 ms. Red arrows indicate the presence of aggregates.

295 ATR-FTIR analysis was performed on the aerogel sponges before and after ethanol annealing to
296 observe the changes in crystallinity (Figure 4). After ethanol annealing, a shift from 1640 cm^{-1} , which
297 corresponds to SF random coils, to 1621 cm^{-1} , which corresponds to SF β -sheets (Hu et al., 2006), was
298 observed for all the sponges with SF. All annealed sponges had a similar β -sheet content of about

299 28%. Without the annealing step, the structure of the sponges collapsed after a few minutes in an
300 aqueous environment.



301
302 Figure 4: FTIR absorbance spectra in the amide I region for aerogel sponges containing SF before and
303 after ethanol annealing. Peak at 1621 cm⁻¹ is typical of the β -sheet conformation of SF, while the
304 peak at 1640 cm⁻¹ corresponds to SF random coils.

305 All aerogel sponges exhibited high porosity, ranging from 82% for the SF-HA-hep sponge to 93% for
306 the SF sponge (Table 1), which inversely correlated with the content in solids used to prepare the
307 starting dispersion (described in section 2.3.1.). The capability of the sponges to uptake water when
308 immersed in NaCl 0.9% (w/v) aqueous solution was of 93% for SF-HA-hep and SF-HA sponges, and
309 95% for HA-PLL sponge (Table 1). Similarities in water uptake may be related to similar porosity and
310 hydrophilicity. Degradation was monitored at 37°C during 7 and 21 days in PBS supplemented with
311 hyaluronidase and heparinase at concentrations that mimic *in vivo* conditions (Correia et al., 2011;
312 Frost, Csoka, Wong, & Stern, 1997) (Figure 5A). HA-PLL sponges were completely degraded after 7

313 days and produced a clear solution, while SF sponges maintained their integrity even after 21 days.
 314 SF-HA and SF-HA-hep showed intermediate degradability. An ideal scaffold for brain regeneration
 315 should be biodegradable to avoid the need for secondary surgery to remove the implant.
 316 Simultaneously, the degradation rate of the scaffold should match the rate of tissue reconstruction
 317 so that the newly formed tissue does not collapse before the regeneration process is complete. Too
 318 slow degrading scaffolds may compromise tissue regeneration, as previously reported for hydrogel-
 319 based brain bioscaffolds (Modo et al., 2019) and SF-based bone scaffolds with tunable degradability
 320 (Park et al., 2010). Therefore, intermediate degradation rates (in the frame of days) have been
 321 pointed out as preferable (Park et al., 2010; Modo et al., 2019). In good agreement with previous
 322 reports, the mass loss in scaffolds composed of HA and SF blends can be attributed to the
 323 degradation of HA (Garcia-Fuentes et al., 2009). Since HA degrades quicker than SF, the matrix has to
 324 be as homogeneous as possible to avoid the generation of scaffolds with distinct HA-rich
 325 compartments susceptible to rapid *in vivo* decomposition.

326 Table 1: Porosity of the aerogel sponges, water uptake in NaCl 0.9% aq. solution, and Young's
 327 modulus after hydration.

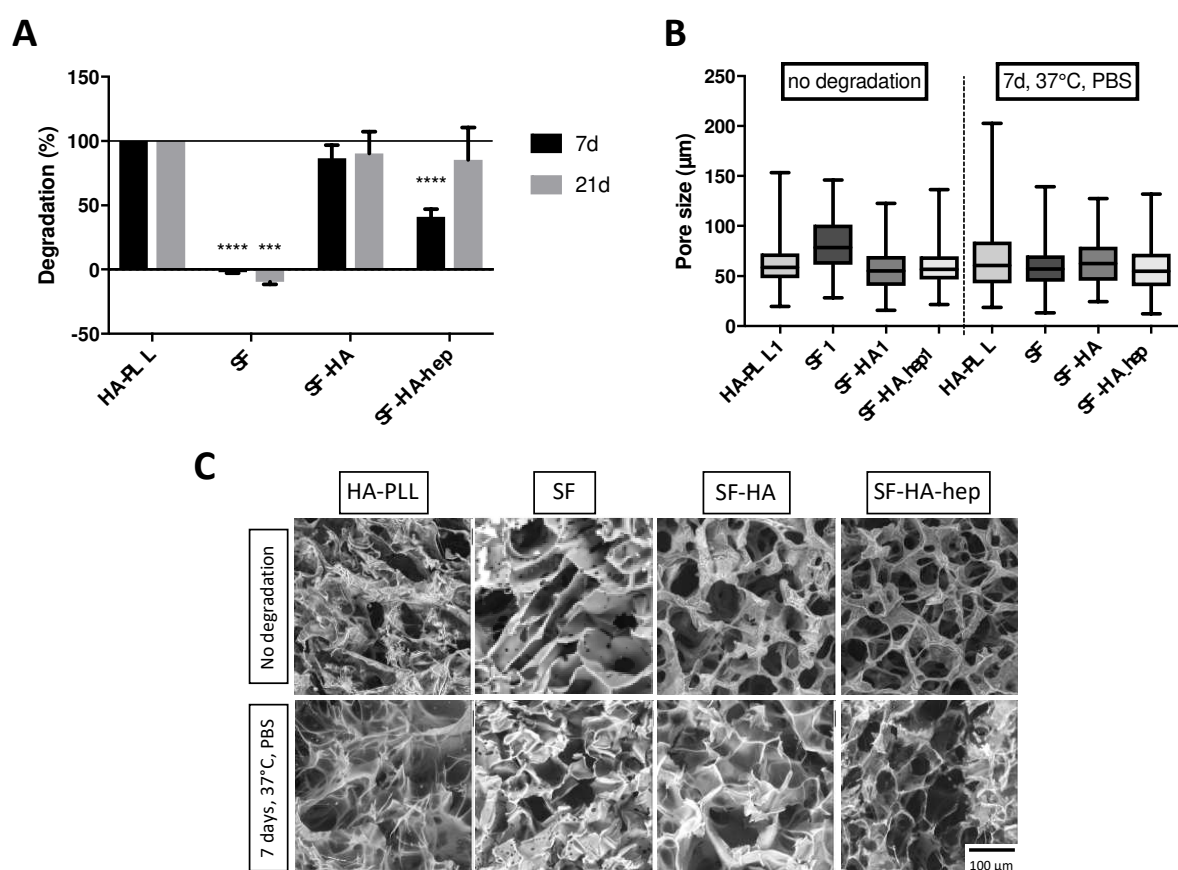
Sponge	Porosity (%)	Water uptake (%)	Young's modulus (kPa)
HA-PLL	91.9 ± 1.9	95.5 ± 0.2	6.0 ± 1.6
SF	92.7 ± 0.6	93.2 ± 0.5	21.7 ± 1.6
SF-HA	87.4 ± 1.0	92.8 ± 2.1	12.5 ± 7.8
SF-HA-hep	82.3 ± 2.2	92.9 ± 2.2	13.1 ± 5.1

328

329 To confirm these results, the morphology of the aerogel sponges was observed using SEM before and
 330 after 7 days in PBS without enzymes at 37°C (Figure 5B-C). These study conditions were selected to
 331 induce intermediate degradation of the HA-PLL sponge. At the end of this study, signs of degradation
 332 in the form of thinner walls and larger pores were clearly visible in HA-PLL sponges. Sponges made of
 333 SF and HA mixtures showed excellent preservation of morphology, confirming that the addition of SF
 334 to the sponges allows for retention of the aerogel structure in aqueous medium. SEM images
 335 revealed good pore connectivity and an average pore size between 30 ± 13 μm for SF-HA and 87 ± 29
 336 μm for SF. The structure of these porous sponges resembles the one of the brain ECM, which has
 337 been described as a foam-like structure (Nicholson & Syková, 1998). The pores were of various sizes
 338 ranging from 20 to 140 μm across the different sponge compositions designed in the present work.
 339 This size range can be considered adequate since the pores are sufficiently small to support 3D cell-
 340 cell contacts, but large enough to allow good diffusion of nutrients, oxygen and bioactive factors for

341 cell survival and growth (Mahumane et al., 2018). The ideal porosity was defined to be around 90%
 342 (Jurga et al., 2011).

343 Texture analysis was performed on hydrated sponges and Young's modulus was calculated (Table 1).
 344 The values of Young's modulus recorded ranged from 6 kPa for HA-PLL to 22 kPa for SF. Sponges
 345 made from mixtures of SF and HA showed intermediate values. The brain has Young's modulus close
 346 to or below 1 kPa (Engler et al., 2006; Nava et al., 2012) although the stiffness varies depending on
 347 the brain region (white matter stiffness is higher and increases with the content in myelin
 348 (Weickenmeier et al., 2016)) and the pathology, increasing in cancer and decreasing in Alzheimer's
 349 disease (Murphy et al., 2011; Wang, Tong, & Yang, 2014). It is still quite challenging to produce
 350 scaffolds that maintain their structural integrity with a Young's modulus as low as the one of the
 351 brain. Most of previous bio-stable scaffolds reported, even those also containing SF, are stiffer than
 352 the sponges developed herein (Wentao et al., 2019). Thus, the aerogel sponges combine stiffness
 353 close to those of brain while they are easy to handle and could be conveniently cut into the desired
 354 dimensions to fit the lesion gap at the intended site of implantation.



355
 356 Figure 5: (A) Degradation of HA-PLL, SF, SF-HA and SF-HA-hep aerogel sponges in PBS with enzymes
 357 (hyaluronidase and heparinase) at 37°C for 7 and 21 days. Paired comparisons were conducted
 358 between degradation of HA-PLL and the others with the same conditions. (B) Average pore size

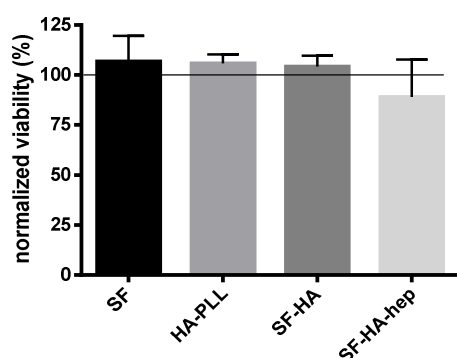
359 before and after incubation of the sponges for 7 days in PBS without enzymes at 37°C. At least 94
360 pores were measured per sponge. (C) SEM images of sponges; scale bar 100 μm . ANOVA:
361 *** $p < 0.001$, **** $p < 0.0001$.

362

363 3.3. Cytocompatibility

364 The cytocompatibility of the sponges was tested, via the indirect and direct contact methods, against
365 the NIH/3T3 mouse fibroblast cell line which is considered as highly sensitive to any scaffold-induced
366 toxicity (Xia et al., 2008). The HA-PLL, SF, SF-HA and SF-HA-hep sponges showed no discernible
367 cytotoxicity with both methods (Figure 6) although the viability slightly decreased for the SF-HA-hep
368 sponge with the direct contact method ($76 \pm 9\%$ viability), but it was not statistically significant.
369 These preliminary tests suggest good cytocompatibility although further studies with several other
370 cell lines and other metabolic activity assays would be required before in vivo testing.

371



372

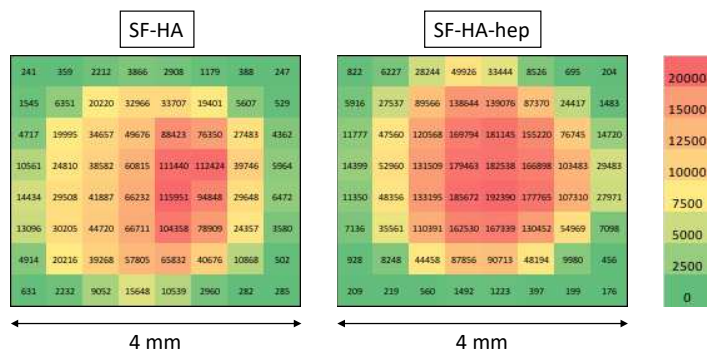
373 Figure 6: Evaluation of the compatibility of SF, HA-PLL, SF-HA and SF-HA-hep aerogel sponges with
374 NIH/3T3 cells.

375 3.4 SDF-1 α uptake and release

376 Since the SF-HA and SF-HA-hep sponges provided a good trade-off between the low rigidity of the
377 HA-PLL sponge and the slow degradation of the SF sponge, they were chosen for the SDF-1 α uptake
378 and release studies.

379 SDF-1 α dissolved in PBS was dropped on the SF-HA and SF-HA-hep sponges. The distribution profile
380 of the protein inside the sponge was visualized using SDF-1 α coupled to AlexaFluor 647. A
381 fluorescence map was subsequently obtained (Figure 7). After 1 h incubation, the protein formed a
382 radial concentration gradient from the center (where the drop was deposited) toward the edges of
383 the sponges. Since it has been described that CXCR4 receptor-bearing cells can migrate in response

384 to an SDF-1 α concentration gradient, the radial distribution of SDF-1 α in the sponge may be
 385 beneficial in facilitating extensive infiltration of stem cells into the sponge for tissue regeneration
 386 (Imitola et al., 2004) as well as for other applications that require the cells to be confined in a matrix
 387 (Najberg et al., 2019).

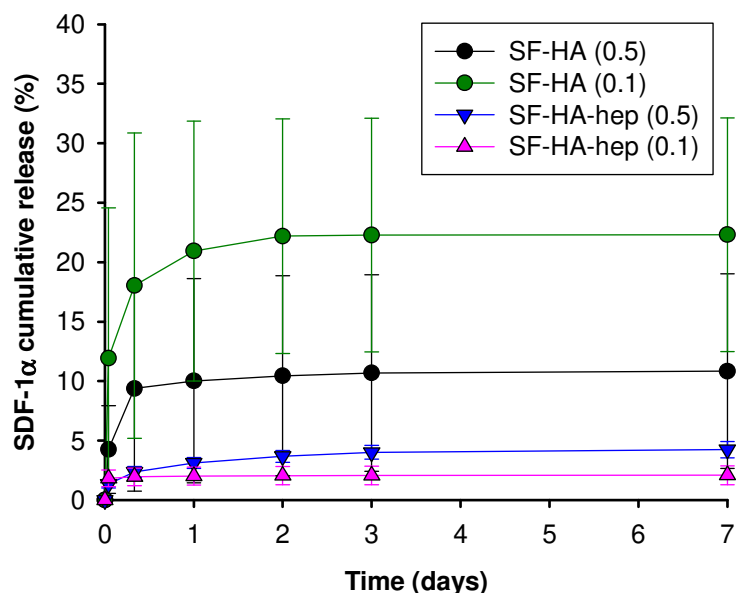


388
 389 Figure 7: Distribution of AF-SDF-1 α in the SF-HA and SF-HA-hep aerogel sponges recorded via
 390 fluorescence detection. Numbers represent the fluorescence intensity at each point. Fluorescence
 391 maps showed here are representative of n=3.

392
 393 For the release studies, SDF-1 α was dropped onto the SF-HA and SF-HA-hep sponges and then the
 394 systems incubated for 1 h or 12 h in order to investigate whether the time of incubation may alter
 395 the interactions strength of the chemokine with the sponge components. SDF-1 α release was
 396 recorded in PBS with enzymes (hyaluronidase and heparinase) at 37°C over 7 days (Figure 8). The
 397 concentration of hyaluronidase was fixed at 0.3 mg/mL, but heparinase was tested at two different
 398 levels, 0.1 and 0.5 U/mL, to cover a wide range of physiological conditions (Correia et al., 2011; Frost,
 399 Csoka, Wong, & Stern, 1997). For the SF-HA-hep sponge, a maximum release of $2.1 \pm 0.8\%$ and $4.0 \pm$
 400 0.6% was reached after 3 days in the release medium prepared with 0.1 and 0.5 U/mL heparinase,
 401 respectively. Therefore, an increase in enzyme promoted the release although the sponges
 402 maintained their integrity, which means that enzyme only altered the heparin moieties. It is expected
 403 that, *in vivo*, the release continues over time in parallel with the degradation of the sponge.
 404 Differently, SF-HA sponge showed a marked burst in the first hours, releasing $22 \pm 10\%$ and $10 \pm 8\%$
 405 after 2 days in the release medium prepared with 0.1 and 0.5 U/mL heparinase, respectively. As
 406 expected these sponges were not sensitive to the increase in heparinase concentration, and the
 407 slower release observed in 0.5 U/mL heparinase medium may be related to the more prolonged
 408 incubation of the sponges with SDF-1 α during loading (12 h instead of 1 h). SDF-1 α is a positively
 409 charged protein at physiological pH (isoelectric point (IEP) ~ 9.6) while HA and SF are negatively
 410 charged (IEP ~ 2.9 and ~ 4.6 respectively) (Foo et al., 2006; Malay, Bayraktar, & Batigün, 2007),

411 allowing the protein to bind to the sponges via electrostatic interactions. In a saline medium, such as
 412 PBS with enzymes, the electrostatic interactions become weaker and therefore the release can
 413 proceed. The dispersion of the release data points may be due to fragmentation of the SF-HA
 414 sponges in pieces of various sizes under the gentle soaking conditions of the release test. Heparin is
 415 known to form a strong complex with SDF-1 α (Sadir, Baleux, Grosdidier, Imberty, & Lortat-Jacob,
 416 2001), which explains the slower release from the SF-HA-hep sponge. Indeed, the strong SDF-1 α
 417 retention capacity displayed by the SF-HA-hep sponge has not been reported before to the best of
 418 our knowledge (Bladergroen et al., 2009; Hu et al., 2018; Krieger et al., 2016; Prokoph et al., 2012;
 419 Purcell, Elser, Mu, Margulies, & Burdick, 2012; Yang et al., 2015).

420 It should be also noted that, compared to other more sophisticated ways of SDF-1 α loading, the SF-
 421 HA-hep sponge allowed for simple, direct loading of relatively large amounts of the protein without
 422 exhibiting any burst effect. A strong retention of SDF-1 α inside the scaffold might be advantageous
 423 for attraction of cells towards the core of the scaffold via haptotaxis as well as cell retention over
 424 time (Monneau, Arenzana-Seisdedos, & Lortat-Jacob, 2016).



425
 426 Figure 8: SDF-1 α cumulative release from the SF-HA and SF-HA-hep aerogel sponges during 7 days in
 427 PBS with enzymes (0.3 mg/mL hyaluronidase and 0.1 or 0.5 U/mL heparinase). Heparinase
 428 concentration in the release medium is indicated in the legend. Sponges tested in the medium with
 429 the highest heparinase concentration (0.5 U/mL) were previously incubated with SDF-1 α for 12 h
 430 (blue and black colors). Other sponges were incubated for 1 h (pink and green colors). All values were
 431 expressed in percentage of the amount of SDF-1 α added to the sponges

432 **4. Conclusion**

433 In summary, cross-linking of mixtures of SF and HA followed by freeze-drying results in aerogel
434 sponges with advantageous properties as soft tissue scaffold compared to those of each biopolymer
435 in separate. The strong autofluorescence of SF allowed for monitoring the homogeneous distribution
436 of the components during sponge preparation and to setup the processing conditions. The SF-HA and
437 SF-HA-hep aerogel sponges exhibited intermediate physicochemical properties (Young's modulus,
438 degradation behaviour) compared to those of SF (~22 kPa; too slow degradation) or HA-PLL(~6 kPa;
439 fast disintegration) sponges. The soft hydrated aerogel sponges showed Young's modulus as low as
440 those of brain tissue, but maintain their integrity for several days in enzyme-containing physiological-
441 like medium. All sponges showed adequate porosity and pore size to be colonized by cells.
442 Nevertheless, the incorporation of heparin resulted to be critical for the hosting and sustained
443 release of the chemokine SDF-1 α . Thus, SF-HA-hep sponges are pointed out as outstanding scaffolds
444 able to uptake relatively large amounts of the protein and prevent burst release for tissue
445 engineering applications.

446

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461 **REFERENCES**

462 Addington, C. P., Heffernan, J. M., Millar-Haskell, C. S., Tucker, E. W., Sirianni, R. W., & Stabenfeldt, S.
463 E. (2015). Enhancing neural stem cell response to SDF-1 α gradients through hyaluronic acid-
464 laminin hydrogels. *Biomaterials*, 72, 11–19. <https://doi.org/10.1016/j.biomaterials.2015.08.041>

465 Bellail, A. C., Hunter, S. B., Brat, D. J., Tan, C., & Van Meir, E. G. (2004). Microregional extracellular
466 matrix heterogeneity in brain modulates glioma cell invasion. *International Journal of*
467 *Biochemistry and Cell Biology*, 36(6), 1046–1069. <https://doi.org/10.1016/j.biocel.2004.01.013>

468 Bladergroen, B. A., Siebum, B., Siebers-Vermeulen, K. G. C., Kuppevelt, T. H. Van, Poot, A. A., Feijen,
469 J., Figdor, C. G., & Torensma, R. (2009). *In vivo* recruitment of hematopoietic cells using stromal
470 cell-derived factor 1 alpha-loaded heparinized three-dimensional collagen scaffolds. *Tissue*
471 *Engineering Part A*, 15(7), 1591–1599.

472 Boni, R., Ali, A., Shavandi, A., & Clarkson, A. N. (2018). Current and novel polymeric biomaterials for
473 neural tissue engineering. *Journal of Biomedical Science*, 25(1), 1–21.
474 <https://doi.org/10.1186/s12929-018-0491-8>

475 Chen, X., Li, W., & Yu, T. (1997). Conformation transition of silk fibroin induced by blending chitosan.
476 *Journal of Polymer Science: Part B: Polymer Physics*, 35, 2293–2296.
477 [https://doi.org/10.1002/\(SICI\)1099-0488\(199710\)35:14<2293::AID-POLB9>3.0.CO;2-X](https://doi.org/10.1002/(SICI)1099-0488(199710)35:14<2293::AID-POLB9>3.0.CO;2-X)

478 Christie, K. J., & Turnley, A. M. (2013). Regulation of endogenous neural stem/progenitor cells for
479 neural repair—factors that promote neurogenesis and gliogenesis in the normal and damaged
480 brain. *Frontiers in Cellular Neuroscience*, 6(January), 1–18.
481 <https://doi.org/10.3389/fncel.2012.00070>

482 Correia, C. R., Moreira-Teixeira, L. S., Moroni, L., Reis, R. L., van Blitterswijk, C. A., Karperien, M., &
483 Mano, J. F. (2011). Chitosan Scaffolds containing hyaluronic acid for cartilage tissue engineering.
484 *Tissue Engineering Part C: Methods*, 17(7), 717–730. <https://doi.org/10.1089/ten.tec.2010.0467>

485 De Souza, S. W., & Dobbing, J. (1971). Cerebral edema in developing brain : i . Normal water and
486 cation content in developing rat brain and postmortem changes. *Experimental Neurology*, 32,
487 431–438.

488 Dehqan Niri, A., Karimi Zarchi, A. A., Ghadiri Harati, P., Salimi, A., & Mujokoro, B. (2019). Tissue
489 engineering scaffolds in the treatment of brain disorders in geriatric patients. *Artificial Organs*,
490 (February), 1–14. <https://doi.org/10.1111/aor.13485>

491 Dutta, D., Fauer, C., Mulleneux, H. L., & Stabenfeldt, S. E. (2015). Tunable controlled release of
492 bioactive SDF-1 α via specific protein interactions within fibrin/nanoparticle composites. *Journal*
493 *of Materials Chemistry B*, 3(40), 7963–7973. <https://doi.org/10.1039/c5tb00935a>

494 Dutta, D., Hickey, K., Salifu, M., Fauer, C., Willingham, C., & Stabenfeldt, S. E. (2017). Spatiotemporal
495 presentation of exogenous SDF-1 with PLGA nanoparticles modulates SDF-1/CXCR4 signaling
496 axis in the rodent cortex. *Biomater. Sci.*, 5(8), 1640–1651.

497 <https://doi.org/10.1039/C7BM00489C>

498 Engler, A. J., Sen, S., Sweeney, H. L., & Discher, D. E. (2006). Matrix elasticity directs stem cell lineage
499 specification. *Cell*, *126*(4), 677–689. <https://doi.org/10.1016/j.cell.2006.06.044>

500 Foo, C. W. P., Bini, E., Hensman, J., Knight, D. P., Lewis, R. V., & Kaplan, D. L. (2006). Role of pH and
501 charge on silk protein assembly in insects and spiders. *Applied Physics A: Materials Science and*
502 *Processing*, *82*(2), 223–233. <https://doi.org/10.1007/s00339-005-3426-7>

503 Frost, G. I., Csoka, T. B., Wong, T., & Stern, R. (1997). Purification, cloning, and expression of human
504 plasma hyaluronidase. *Biochemical and Biophysical Research Communications*, *236*, 10–15.
505 <https://doi.org/10.1166/jnn.2007.427>

506 Führmann, T., Obermeyer, J., Tator, C. H., & Shoichet, M. S. (2015). Click-crosslinked injectable
507 hyaluronic acid hydrogel is safe and biocompatible in the intrathecal space for ultimate use in
508 regenerative strategies of the injured spinal cord. *Methods*, *84*, 60–69.
509 <https://doi.org/10.1016/j.ymeth.2015.03.023>

510 García-González, C. A., Budtova, T., Durães, L., Erkey, C., Del Gaudio, P., Gurikov, P., Koebel, M.,
511 Liebner, F., Neagu, M., & Smirnova, I. (2019). An opinion paper on aerogels for biomedical and
512 environmental applications. *Molecules*, *24*, 1815. <https://doi.org/10.3390/molecules24091815>

513 Garcia-Fuentes, M., Giger, E., Meinel, L., & Merkle, H. P. (2008). The effect of hyaluronic acid on silk
514 fibroin conformation. *Biomaterials*, *29*(6), 633–642.
515 <https://doi.org/10.1016/j.biomaterials.2007.10.024>

516 Garcia-Fuentes, M., Meinel, A. J., Hilbe, M., Meinel, L., & Merkle, H. P. (2009). Silk fibroin/hyaluronan
517 scaffolds for human mesenchymal stem cell culture in tissue engineering. *Biomaterials*, *30*(28),
518 5068–5076. <https://doi.org/10.1016/j.biomaterials.2009.06.008>

519 González-Nieto, D., Fernández-García, L., Pérez-Rigueiro, J., Guinea, G. V., & Panetsos, F. (2018).
520 Hydrogels-assisted cell engraftment for repairing the stroke-damaged brain: Chimera or reality.
521 *Polymers*, *10*, 184. <https://doi.org/10.3390/polym10020184>

522 Haji Mansor, M., Najberg, M., Contini, A., Alvarez-Lorenzo, C., Garcion, E., Jérôme, C., & Boury, F.
523 (2018). Development of a non-toxic and non-denaturing formulation process for encapsulation
524 of SDF-1 α into PLGA/PEG-PLGA nanoparticles to achieve sustained release. *European Journal of*
525 *Pharmaceutics and Biopharmaceutics*, *125*(November 2017), 38–50.
526 <https://doi.org/10.1016/j.ejpb.2017.12.020>

527 Haring, A. P., Thompson, E. G., Tong, Y., Laheri, S., Cesewski, E., Sontheimer, H., & Johnson, B. N.
528 (2019). Process- and bio-inspired hydrogels for 3D bioprinting of soft free-standing neural and

529 glial tissues. *Biofabrication*, 11(2). <https://doi.org/10.1088/1758-5090/ab02c9>

530 Hu, X., Kaplan, D., & Cebe, P. (2006). Determining beta-sheet crystallinity in fibrous proteins by
531 thermal analysis and infrared spectroscopy. *Macromolecules*, 39(18), 6161–6170.
532 <https://doi.org/10.1021/ma0610109>

533 Hu, X., Shmelev, K., Sun, L., Gil, E. S., Park, S. H., Cebe, P., & Kaplan, D. L. (2011). Regulation of silk
534 material structure by temperature-controlled water vapor annealing. *Biomacromolecules*, 12(5),
535 1686–1696. <https://doi.org/10.1021/bm200062a>

536 Hu, Y., Ran, J., Zheng, Z., Jin, Z., Chen, X., Yin, Z., Tang, C., Chen, Y., Huang, J., Le, H., Yan, R., Zhu, T.,
537 Wang, J., Lin, J., Xu, K., Zhou, Y., Zhang, W., Cai, Y., Dominique, P., Heng, B. C., Chen, W., Shen,
538 W., & Ouyang, H. W. (2018). Exogenous stromal derived factor-1 releasing silk scaffold
539 combined with intra-articular injection of progenitor cells promotes bone-ligament-bone
540 regeneration. *Acta Biomaterialia*, 71, 168–183. <https://doi.org/10.1016/j.actbio.2018.02.019>

541 Imitola, J., Raddassi, K., Park, K. I., Mueller, F.-J., Nieto, M., Teng, Y. D., Frenkel, D., Li, J., Sidman, R.L.,
542 Walsh, C. A., Snyder, E. Y., Khoury, S. J. (2004). Directed migration of neural stem cells to sites of
543 CNS injury by the stromal cell-derived factor 1 /CXC chemokine receptor 4 pathway.
544 *Proceedings of the National Academy of Sciences*, 101(52), 18117–18122.
545 <https://doi.org/10.1073/pnas.0408258102>

546 Jian, W. H., Wang, H. C., Kuan, C. H., Chen, M. H., Wu, H. C., Sun, J. S., & Wang, T. W. (2018).
547 Glycosaminoglycan-based hybrid hydrogel encapsulated with polyelectrolyte complex
548 nanoparticles for endogenous stem cell regulation in central nervous system regeneration.
549 *Biomaterials*, 174, 17–30. <https://doi.org/10.1016/j.biomaterials.2018.05.009>

550 Jurga, M., Dainiak, M. B., Sarnowska, A., Jablonska, A., Tripathi, A., Plieva, F. M., ... McGuckin, C. P.
551 (2011). The performance of laminin-containing cryogel scaffolds in neural tissue regeneration.
552 *Biomaterials*, 32(13), 3423–3434. <https://doi.org/10.1016/j.biomaterials.2011.01.049>

553 Krieger, J. R., Ogle, M. E., McFaline-Figueroa, J., Segar, C. E., Temenoff, J. S., & Botchwey, E. A. (2016).
554 Spatially localized recruitment of anti-inflammatory monocytes by SDF-1 α -releasing hydrogels
555 enhances microvascular network remodeling. *Biomaterials*, 77, 280–290.
556 <https://doi.org/10.1016/j.biomaterials.2015.10.045>

557 Kuo, J. W., Swarm, D. A., & Prestwich, G. D. (1991). Chemical modification of hyaluronic acid by
558 carbodiimides. *Bioconjugate Chemistry*, 2(4), 232–241. <https://doi.org/10.1021/bc00010a007>

559 Lassoued, N., Delarue, J., Launay, B., & Michon, C. (2008). Baked product texture: Correlations
560 between instrumental and sensory characterization using Flash Profile. *Journal of Cereal*

561 *Science*, 48(1), 133–143. <https://doi.org/10.1016/j.jcs.2007.08.014>

562 Lau, T. T., & Wang, D. A. (2011). Stromal cell-derived factor-1 (SDF-1): Homing factor for engineered
563 regenerative medicine. *Expert Opinion on Biological Therapy*, 11(2), 189–197.
564 <https://doi.org/10.1517/14712598.2011.546338>

565 Mahumane, G. D., Kumar, P., Du Toit, L. C., Choonara, Y. E., & Pillay, V. (2018). 3D scaffolds for brain
566 tissue regeneration: Architectural challenges. *Biomaterials Science*, 6(11), 2812–2837.
567 <https://doi.org/10.1039/c8bm00422f>

568 Malay, Ö., Bayraktar, O., & Batigün, A. (2007). Complex coacervation of silk fibroin and hyaluronic
569 acid. *International Journal of Biological Macromolecules*, 40(4), 387–393.
570 <https://doi.org/10.1016/j.ijbiomac.2006.09.017>

571 Martín-Martín, Y., Fernández-García, L., Sanchez-Rebato, M. H., Marí-Buyé, N., Rojo, F. J., Pérez-
572 Rigueiro, J., Ramos, M., Guinea, G.V., Panetsos, F., & González-Nieto, D. (2019). Evaluation of
573 neurosecretome from mesenchymal stem cells encapsulated in silk fibroin hydrogels. *Scientific*
574 *Reports*, 9(1), 1–15. <https://doi.org/10.1038/s41598-019-45238-4>

575 Mazia, D., Schatten, G., & Sale, W. (1975). Adhesion of cells to surfaces coated with polystyrene.
576 Applications to electron microscopy. *Journal of Cell Biology*, 66(3), 198–200.

577 Mobini, S., Taghizadeh-Jahed, M., Khanmohammadi, M., Moshiri, A., Naderi, M.-M., Heidari-Vala, H.,
578 Ashrafi Helan, J., Khanjani, S., Springer, A., Akhondi, M. M., & Kazemnejad, S. (2016).
579 Comparative evaluation of *in vivo* biocompatibility and biodegradability of regenerated silk
580 scaffolds reinforced with/without natural silk fibers. *Journal of Biomaterials Applications*, 30(6),
581 793–809. <https://doi.org/10.1177/0885328215601925>

582 Modo, M. (2019). Bioscaffold-induced brain tissue regeneration. *Frontiers in Neuroscience*, 13, 1156.
583 <https://doi.org/10.3389/fnins.2019.01156>

584 Monneau, Y., Arenzana-Seisdedos, F., & Lortat-Jacob, H. (2016). The sweet spot: how GAGs help
585 chemokines guide migrating cells. *Journal of Leukocyte Biology*, 99(6), 935–953.
586 <https://doi.org/10.1189/jlb.3mr0915-440r>

587 Murphy, M. C., Huston, J., Jack, C. R., Glaser, K. J., Manduca, A., Felmlee, J. P., & Ehman, R. L. (2011).
588 Decreased brain stiffness in Alzheimer’s disease determined by magnetic resonance
589 elastography. *Journal of Magnetic Resonance Imaging*, 34(3), 494–498.
590 <https://doi.org/10.1002/jmri.22707>

591 Najberg, M., Haji Mansor, M., Boury, F., Alvarez-Lorenzo, C., & Garcion, E. (2019). Reversing the
592 tumor target: establishment of a tumor trap. *Frontiers in Pharmacology*, 10(August), 1–12.

593 <https://doi.org/10.3389/fphar.2019.00887>

594 Nava, M. M., Raimondi, M. T., & Pietrabissa, R. (2012). Controlling self-renewal and differentiation of
595 stem cells via mechanical cues. *Journal of Biomedicine and Biotechnology*, 2012, 1–12.
596 <https://doi.org/10.1155/2012/797410>

597 Nicholson, C., & Syková, E. (1998). Extracellular space structure revealed by diffusion analysis. *Trends*
598 *in Neurosciences*, 21, 207–215. [https://doi.org/10.1016/S0166-2236\(98\)01261-2](https://doi.org/10.1016/S0166-2236(98)01261-2)

599 Nultsch, K., & Germershaus, O. (2017). Silk fibroin degumming affects scaffold structure and release
600 of macromolecular drugs. *European Journal of Pharmaceutical Sciences*, 106(June), 254–261.
601 <https://doi.org/10.1016/j.ejps.2017.06.012>

602 Orive, G., Anitua, E., Pedraz, J. L., & Emerich, D. F. (2009). Biomaterials for promoting brain
603 protection, repair and regeneration. *Nature Reviews Neuroscience*, 10(9), 682–692.
604 <https://doi.org/10.1038/nrn2685>

605 Park, S.H., Gil, E.S., Shi, H., Kim, H.J., Lee, K., Kaplan, D.L. (2010). Relationships between degradability
606 of silk scaffolds and osteogenesis. *Biomaterials*, 31(24), 6162–6172.
607 <https://doi.org/10.1016/j.biomaterials.2010.04.028>

608 Prokoph, S., Chavakis, E., Levental, K. R., Zieris, A., Freudenberg, U., Dimmeler, S., & Werner, C.
609 (2012). Sustained delivery of SDF-1 α from heparin-based hydrogels to attract circulating
610 pro-angiogenic cells. *Biomaterials*, 33(19), 4792–4800.
611 <https://doi.org/10.1016/j.biomaterials.2012.03.039>

612 Purcell, B. P., Elser, J. A., Mu, A., Margulies, K. B., & Burdick, J. A. (2012). Synergistic effects of SDF-1 α
613 chemokine and hyaluronic acid release from degradable hydrogels on directing bone marrow
614 derived cell homing to the myocardium. *Biomaterials*, 33(31), 7849–7857.
615 <https://doi.org/10.1016/j.biomaterials.2012.07.005>

616 Rabbany, S. Y., Pastore, J., Yamamoto, M., Miller, T., Rafii, S., Aras, R., & Penn, M. (2010). Continuous
617 delivery of stromal cell-derived factor-1 from alginate scaffolds accelerates wound healing. *Cell*
618 *Transplantation*, 19(4), 399–408. <https://doi.org/10.3727/096368909X481782>

619 Rockwood, D. N., Preda, R. C., Yücel, T., Wang, X., Lovett, M. L., & Kaplan, D. L. (2011). Materials
620 fabrication from *Bombyx mori* silk fibroin. *Nature Protocols*, 6(10), 1612–1631.
621 <https://doi.org/10.1038/nprot.2011.379>

622 Rouslahti, E. (1996). Brain extracellular matrix. *Glycobiology*, 6(5), 489–492.

623 Sadir, R., Baleux, F., Grosdidier, A., Imbert, A., & Lortat-Jacob, H. (2001). Characterization of the

624 stromal cell-derived factor-1alpha-heparin complex. *J Biol Chem*, 276(11), 8288–8296.
625 <https://doi.org/10.1074/jbc.M008110200>

626 Seyedhassantehrani, N., Li, Y., & Yao, L. (2016). Dynamic behaviors of astrocytes in chemically
627 modified fibrin and collagen hydrogels. *Integrative Biology (United Kingdom)*, 8(5), 624–634.
628 <https://doi.org/10.1039/c6ib00003g>

629 Thurber, A. E., Omenetto, F. G., & Kaplan, D. L. (2015). In vivo bioresponses to silk proteins.
630 *Biomaterials*, 71(2015), 145–157. <https://doi.org/10.1016/j.biomaterials.2015.08.039>

631 Vepari, C., & Kaplan, D. L. (2012). Silk as biomaterial. *Progress in Polymer Science*, 100(2), 130–134.
632 <https://doi.org/10.1016/j.pestbp.2011.02.012>.

633 Wang, C., Tong, X., & Yang, F. (2014). Bioengineered 3D Brain tumor model to elucidate the effects of
634 matrix stiffness on glioblastoma cell behavior using peg-based hydrogels. *Molecular
635 Pharmaceutics*, 11, 2115–2125. <https://doi.org/10.1021/mp5000828> |

636 Weickenmeier, J., de Rooij, R., Budday, S., Steinmann, P., Ovaert, T. C., & Kuhl, E. (2016). Brain
637 stiffness increases with myelin content. *Acta Biomaterialia*, 42, 265–272.
638 <https://doi.org/10.1016/j.actbio.2016.07.040>

639 Wentao, Z., Ya'nan, H., Jian, L., Kaipeng, B., Peng, S., Yu, Z., Peng, Z., Huanxiang, Z., Feng, Z., & Yixin,
640 S. (2019). In vitro biocompatibility study of a water-rinsed biomimetic silk porous scaffold with
641 olfactory ensheathing cells. *International Journal of Biological Macromolecules*, 125, 526–533.
642 <https://doi.org/10.1016/j.ijbiomac.2018.11.029>

643 Xia, M., Huang, R., Witt, K. L., Southall, N., Fostel, J., Cho, M. H., Jadhav, A., Smith, C. S., Inglese, J.,
644 Portier, C. J., Tice, R. R., & Austin, C. P. (2008). Compound cytotoxicity profiling using
645 quantitative high-throughput screening. *Environmental Health Perspectives*, 116(3), 284–291.
646 <https://doi.org/10.1289/ehp.10727>

647 Yamada, H., Nakao, H., Takasu, Y., & Tsubouchi, K. (2001). Preparation of undegraded native
648 molecular fibroin solution from silkworm cocoons. *Materials Science and Engineering: C*, 14(1),
649 41–46. [https://doi.org/10.1016/S0928-4931\(01\)00207-7](https://doi.org/10.1016/S0928-4931(01)00207-7)

650 Yang, J. W., Zhang, Y. F., Wan, C. Y., Sun, Z. Y., Nie, S., Jian, S. J., Zhang, L., Song, G. T., & Chen, Z.
651 (2015). Autophagy in SDF-1a-mediated DPSC migration and pulp regeneration. *Biomaterials*, 44,
652 11–23. <https://doi.org/10.1016/j.biomaterials.2014.12.006>

653 Zhao, W., Jin, K., Li, J., Qiu, X., & Li, S. (2017). Delivery of stromal cell-derived factor 1 α for *in situ*
654 tissue regeneration. *Journal of Biological Engineering*, 11(1), 1–12.
655 <https://doi.org/10.1186/s13036-017-0058-3>